

# Nanomechanical properties of dehydrated enamel surface after bleaching treatment.

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## Abstract

The abstract should summarize the content of the paper. Try to keep the abstract below 150 words. The aim of the present study was to investigate the effects of bleaching procedures of a high concentration of hydrogen peroxide (HP) on dehydrated enamel surface. Flat enamel samples embedded in epoxy resin were dehydrated and treated with 35% HP of photo-initiating bleaching system. Mechanical properties such as nanohardness ( $H$ ), elastic modulus ( $E$ ) and the stiffness ( $S$ ) of the enamel were analyzed by nanoindentation method. Residual indentation of samples was recorded by the Atomic Force Microscopy (AFM) Nano Vision system attached to the nanoindenter instrument. Surfaces of treated enamel tissue, studied under scanning electron microscopy (SEM), X-ray diffraction and AFM determinate the crystallographic spectrum. The  $H$ ,  $E$  and  $S$  as well as surface roughness of enamel surface were significantly decreased by the effects of the bleaching solution treatment ( $H$  and  $E < 50\%$ ) the effects of demineralization observed thought changes on minerals quantities by the absence of contact with saliva.

**Keywords:** Enamel, bleaching treatment, mechanical properties and nanoindentation.

## 1. Introduction

By aesthetic reasons bleaching procedures are performance on human teeth surface structured by enamel rods. Enamel is a highly mineralized structure consisting of approximately 96% mineral and 4% organic material and water. The fundamental units of enamel are carbonated hydroxyapatite crystals [1]. Previous investigations have been focused on the mechanical characterization of whole teeth and larger sizes of enamel samples at microscopic scale, reporting elastic modulus within the range of 74–130 GPa; nanoindentation revealed elastic modulus of 120 GPa [2]. Nanoindentation technology has been recently introduced for the evaluation of mechanical properties of human enamel after bleaching treatments, evidencing changes of mechanical behavior of

enamel at nanoscale on biomimetic materials [3]. Different studies indented the enamel with a Berkovich indenter under close loop mode, loading (loading rate 0.43 mN/s) and unloading with 20 data points and a maximum force of 15 mN to calculates the hardness and elastic modulus from the load–displacement curve based on the power-law fitting method for unloading part of the load–displacement curve [4]. Enamel in original states, contains mainly calcium phosphate salts in shapes of larges hexagonal hydroxyapatite crystals, within averages of hardness ( $H$ ) and Young's modulus ( $E$ ) of.  $H > 6\text{GPa}$  and  $E > 115\text{GPa}$  [5]. The exact mechanism by which hydrogen peroxide affects the enamel has yet to be fully elucidated. After, bleached with 30% hydrogen peroxide on enamel surface of premolars, mean hardness decreased between 13 to 32%, while the mean of Young's modulus decreased by 18–32% [6], determinates the wear behavior of the enamel was by employing nano scratch test, while studied hardness and elastic modulus were by nanoindentation with Berkovich tip. Atomic force microscopy (AFM) combined with nanoindentation techniques, can also be used to observe changes on the surface roughness on enamel surface as well as mechanical properties at nanometric level [7], AFM nanoindentation in combination with electron microscopy analysis it has been use to analyzed demineralization on enamel [5], [8], also to measure the stiffness at nano scale on biological specimens [3]. Bleaching is closely correlated with variations on the amounts of  $\text{Ca}^{2+}$  extracted from the enamel, after bleaching with 30% hydrogen peroxide for 1 h, there were no changes in amounts of  $\text{Ca}^{2+}$  extracted because specimens were stored in artificial saliva [9], although enamel surface treated with 38% hydrogen peroxide for 15 minutes, did not reduce the bonding properties after orthodontic brackets position, but generate changes on the enamel surface [10]. Some authors assert that hydrogen peroxide does not cause any damages on enamel surface [3], [7]. Garrido *et al*, mention that after

initial increase, the enamel exhibited a recovery of properties when it was stored in artificial saliva [11]. The mean nanohardness of the original enamel rods was calculated to be 5.31 to 7.15GPa, while the reduced elastic modulus was determined to be 86.38 to 70.96 GPa on enamel rods, affected by surface treatment [12]. Recent studios propose a reasonable doubt about bleaching therapies may have a negative effect on physical properties of the enamel [13], [14]. The present study was to investigate the effects of bleaching procedures of a high concentration of hydrogen peroxide (HP) on dehydrated enamel surface. Flat enamel samples embedded in epoxy resin were dehydrated and treated with 35% HP of photo-initiating bleaching system. Mechanical properties such as nanohardness ( $H$ ), elastic modulus ( $E$ ) and the stiffness ( $S$ ) of the enamel were analyzed by nanoindentation method using a Berkovich tip. On the other hand, to determinate the Hertzian behavior and the yield point by the pop-ins transition through the load-penetration behavior curve were used the sphere-conical tip. Surfaces of treated enamel tissue, studied under scanning electron microscopy, X-ray diffraction and AFM determinate the crystallographic spectrum and the morphology of the structured before and after of bleaching treatment.

## 2. Materials and methods

### 2.1 Enamel samples preparation.

Four Flat enamel samples in flake shape embedded in epoxy resin were prepared from vestibular area of two human premolar teeth (specimen 1 and specimen 2) and stored in dry environment for one year. One group of enamel samples from the same specimen, were treated with 35% hydrogen peroxide and receive 30 minutes of fotocure trough halogen high intensity light exposition and finally washed with water simply alone, while the other untreated samples specimens constituted the control group. After bleaching samples were stored on dry environment.

### 2.2 Morphology, structural and Nanomechanical characterization.

Standard procedure Energy Dispersive Spectroscopy (EDS) by a scanning electron microscope Jeol model was used for identifying and quantifying elemental of Ca, C, O, F, Cl and P principally and morphology of the exposed tissues treated with 35% HP and the controls. The crystalline structure of the enamel before and after bleaching treatment was analyzed using a Panalytical X-Pert system. The patterns were obtained using Cu  $K\alpha$  radiation at 40 kV and 35 mA. The diffracted beam path included a graphite flat crystal monochromator. The

grazing incidence angle was fixed at  $0.5^\circ$ , whereas the scanning angle  $2\theta$  was varied between  $20$  and  $75^\circ$ , with a step size of  $0.02^\circ$ .

On the other hand, before and 72 hrs after bleaching treatment, mechanical properties such as hardness and elastic modulus were evaluated by means of nanoindentation, employing a Nano Indenter G200 coupled with a DCM II head. The equipment was calibrated by using a standard fused silica sample. Tests parameters were the following; the constants of area function were  $C_0=24.05$ ,  $C_1=-178.33$ ,  $C_2=6724.30$ ,  $C_3=-24407.23$ , and  $C_5=18701.80$ . Berkovich diamond indenter with a tip radius of  $20 \pm 5$  nm, maximum load of 35 mN, strain rate of  $0.05 \text{ s}^{-1}$ , and harmonic displacement and frequency of 1 nm and 75 Hz, passion's coefficient of  $\nu=0.25$  respectively. 90 indentations tests were evaluated in each sample. Residual indentation of samples was recorded by the AFM Nano Vision system attached to the nanoindenter system. During indentation, a curve describing the relationship between load ( $P$ ) and displacement ( $h$ ) is continuously monitored and recorded as illustrated in Fig. (5). On the other hand, with propose to measurement the nano-mechanical properties of the enamel before and after bleaching treatment was used the Oliver and Pharr method with controlled cycles [15]. The basic analysis of nanoindentation load-displacement curve ( $P-h$ ) was established based on the elastic contact theory given by Sneddon [16] and Doerner [17]. The Sneddon equation's following was used to determinate the elastic modulus.

$$S = 2\beta \sqrt{\frac{A}{\pi}} E_r \quad (1)$$

where  $\beta$  is a constant that depends on the geometry of the indenter ( $\beta = 1.034$  for a Berkovich indenter),  $E_r$  is the reduced elastic modulus, which accounts for the fact that elastic deformation occurs in both the sample and the indenter and  $A$  is the contact area that is function of the penetration depth or displacement ( $h$ )[15]. The elastic modulus,  $E$  can be calculated by considering the compliance of the specimen and the indenter tip combined in series,

$$E = \frac{1-\nu^2}{\left(\frac{1}{E_r} - \frac{1-\nu_i^2}{E_i}\right)} \quad (2)$$

where  $E_i$ ,  $E$  and  $\nu_i$ ,  $\nu$  are elastic modulus and Poisson's ratio of diamond indenter and specimen respectively. For the diamond indenter  $E_i = 1140$  GPa and  $\nu_i = 0.07$  are used (G200 Aguiet manual, Agilent technology USA).

The hardness was calculated by the following equation

$$H = P_{\max}/A(h) \quad (3)$$

Where  $H$  is the hardness,  $P_{max}$  is the maximum load and  $A(h)$  is the contact area between the sample and tip indenter, that is function of the penetration depth.

On the other hand, there is a possibility to use indentation with sphere-conical tip to probe the Hertzian behavior and yield points of a material before and after bleaching treatment, but it must be pointed out that measured indentation yield points are different from uniaxial mechanical tests due to the complex 3D-stress – distribution around the indent. A spherical indenter in contact with a specimen surface is shown in (Fig. 7 a)). The indentation area,  $A$  can be related to the indentation radius,  $a$ , or to the indenter radius,  $R=1000nm$  and the contact depth,  $h_c$ :

$$A = \pi a^2 = \pi(2Rh_c - h_c^2) \quad (4)$$

classical Hertzian equation for elastic contact of an elastically isotropic material denotes [18].

$$a = \left( \frac{3PR}{4E_r} \right)^{1/3} \quad (5)$$

where  $a$  contact radius,  $P$  is the applied load,  $R$  is the indenter radius and  $E_r$  is the reduced elastic modulus.

### 3. Results and discussion

In the Figure 1 (a) and (b) show the morphology of the specimen 1 before and after bleaching treatment respectively.

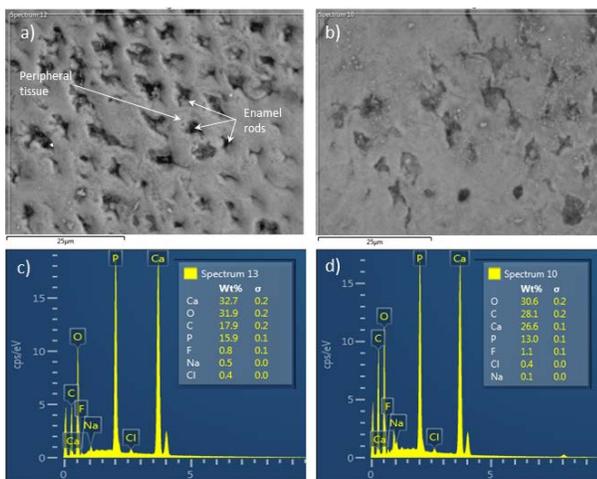


Fig. 1 Specimen1: (a), (b) Morphology and (c), (d) EDS before and after bleaching treatment respectively.

In this figure we can clearly observed the surfaces of treated enamel tissue manifest erosion. On the other hand, in the figure 1 c) and d) by the analysis EDS was possible identifying and quantifying elemental of Ca, C, O, F, Cl and P principally. In this analysis is observed that Ca decrease of the 32.7 to 26.6 Wt %. Moreover, In the Figure 2 that corresponding at the specimen 2 show the same behavior the evidence of the Ca decrease of the 33.7 to 31.2 Wt %.

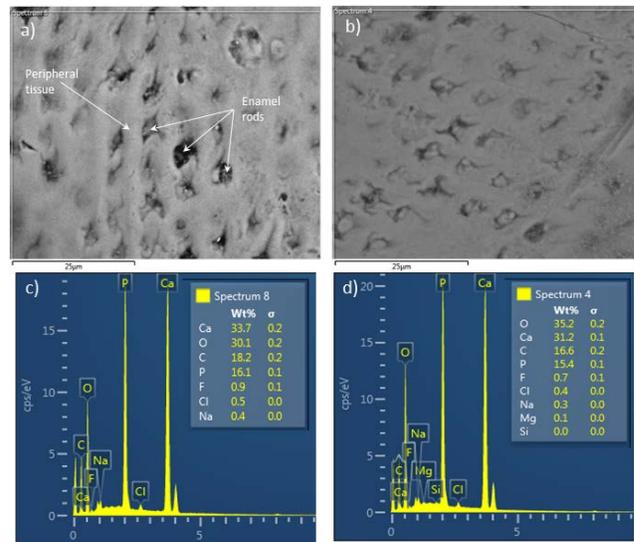


Fig. 2 Specimen 2: (a), (b) Morphology and (c), (d) EDS before and after bleaching treatment respectively.

On the other hand, the figure 3a) and 3b) show the images AFM in contact mode of the enamel (in peripheral tissue region) before and after bleaching treatment respectively. The fig. 3a) denotes small dressings pollutants of tartar material on the surface of hydroxyapatite arrangements. On the zoom images on Peritubular dentin region images it is possible to observe at high magnification, the erosive effect of the bleaching treatment, as well as the removal of contaminants from the enamel surface. Moreover, the Figure 4 shows the results of the XRD analysis of the representative specimen 1 before and after of bleaching treatment a) and b) respectively. In this figure we can clearly observed the decalcification because calcite lost after the bleaching treatment, it correspondent to lost characteristic peak (0 2 0) at ~22 of  $2\theta$  and generally intensities. With this results we can evidence an erosive damage of the enamel tissue.

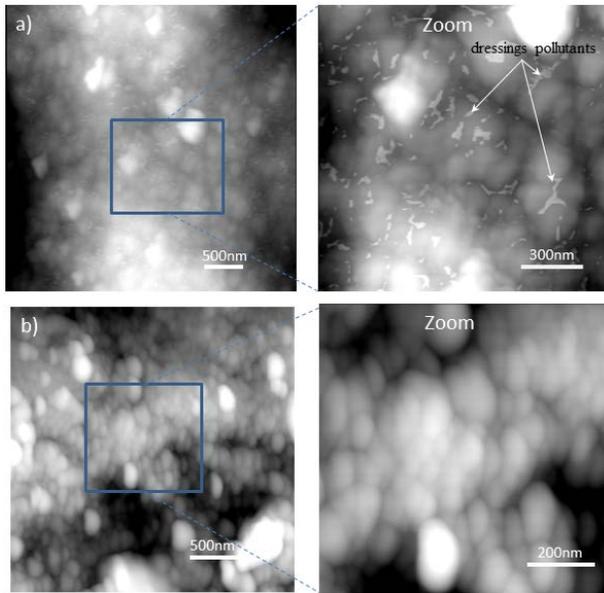


Fig. 3 a) and b) AFM imagines in contact mode of the enamel before and after bleaching treatment respectively, as well as zoom areas in Enamel tissue and interrod crystals.

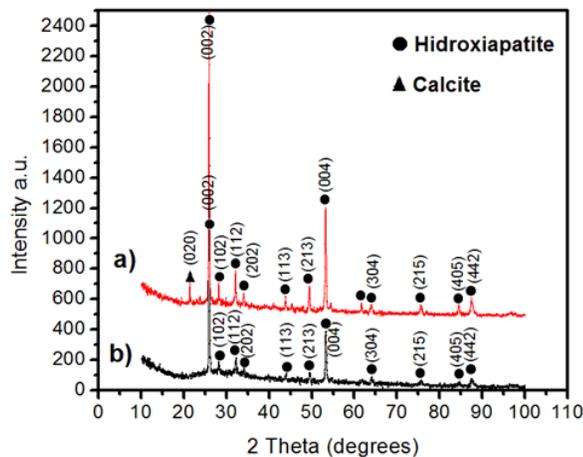


Fig. 4 XRD of representative specimen 1 before and after bleaching treatment a) and b) respectively.

On the other hand, the mechanical properties were evaluated by nanoindentation with tip Bekovich before and after treatment bleaching by the application of 35% hydrogen peroxide. Figure 5 shows the average representative's load versus penetration depth before and after bleaching treatment; a) specimen 1 and b) specimen 2. In the figure 5a) corresponding to the specimen 2 is clearly seen that mechanical behavior is different due to at the

average values residual penetration or final depth ( $h_f=h_r$ ),  $h_r = 56 \pm 4\text{nm}$  and  $200 \pm 5\text{nm}$  before and after treatment bleaching respectively.

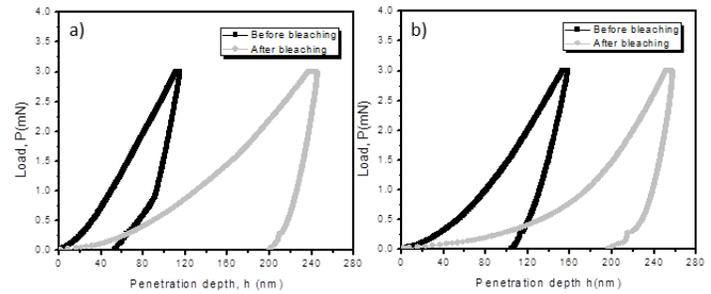


Fig. 5 Characteristic load-penetration depth curves in nanoindentation before and after bleaching treatment a) specimen 1 and b) specimen 2.

On the other hand, surface roughness observed by AFM images of the nanoindent impressions in the prisms of the surface. The figure 6 show optical image a), and b), c) AFM images in contact mode take before and after indentation test. In the figure c) show the representative Berkovich residual indentation area.

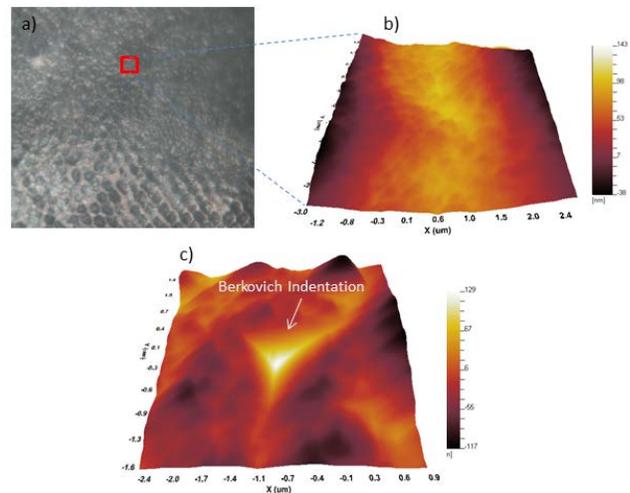


Fig. 6.- a) Optical image, b) and c) AFM images in contact mode taken before and after indentation test with Berkovich tip.

Respected the Figure 5 b) is the same behavior, the values of the residuals penetration depth are  $107 \pm 2\text{nm}$   $199 \pm 3\text{nm}$ . According to the results, nanomechanical behavior of enamel was affected by the application of 35% hydrogen peroxide. For example, in the specimen 1 nanohardness

and elastic modulus registered variations aimed at the reduction by effects of bleaching in averages values of  $H=9.2 \pm 02$  to  $H=3.0 \pm 03$ ,  $E=270 \pm 3$  to  $E=105 \pm 2$  respectively. There were significant decreases in the mean hardness (<60%) and Young's modulus (<50%). Mechanical properties of the two specimens before and after treatment bleaching such as elastic modulus, hardness and stiffness values showed in table 1.

Table 1.- Mechanical properties such as nanohardness (H), elastic modulus (E) and the stiffness (S) of the enamel before and after bleaching treatment evaluated by nanoindentation.

Specimen	Bleaching 35% HP	Hardness (GPa)	Elastic modulus (GPa)	Stiffness (N/m)
1	Not treated	9.4±0.2	270±3	150962±234
	Treated	3.0±0.3	105±2	139622±154
2	Not treated	8.1±0.5	224±4	150281±332
	Treated	3.5±0.4	128±2	120339±247

On the other hand, with the aim to observed the elastic to plastic transition and Hertzian behavior of the enamel before and after bleaching treated was used a share-conical tip. In the figure 7a) show the SEM image of the sphere-conical tip and the zoom images show perfectly the tip radii, which is of  $R=1000$  nm. The figure 7b) show the load-displacement curves before and after bleaching representative specimen 1, as well as the only elastic or Hertzian behavior to the without bleaching treatment. In this case we can see that there is not residual displacement or final depth, ( $h_f=h_r=0$ ) due to the load is not enough to exceed the transition elastic to plastic. It is observed by the pop-ins events in the load-displacement curves at 290µN and 220 µN of load to before and after bleaching treatment respectively.

Still there is some disagreement in the literature about the actual effect of bleaching agents on the mechanical properties of enamel, one of the driving forces of this work is to contribute for clarify this situation. Taking into account that this study was conducted in dehydrated enamel, our results were higher but consistent respect to others authors such as [11],[19]. Averages close to  $H > 6$  GPa and  $E > 115$  GPa reported, employing similar nanoindentation techniques, also [2] reported elastic modulus within the range of 74–130 GPa on untreated enamel, while [20], reported  $E > 114$  GPa for natural enamel in white demineralized by spot lesions. Mechanical properties, chemistry and microstructure of enamel observed altered, showing a correlation with reductions of CaO wt.%, these results support the fact that

the mechanical properties of enamel dependent on its degree of mineralization ( $CaO/P_2O_5$ ).

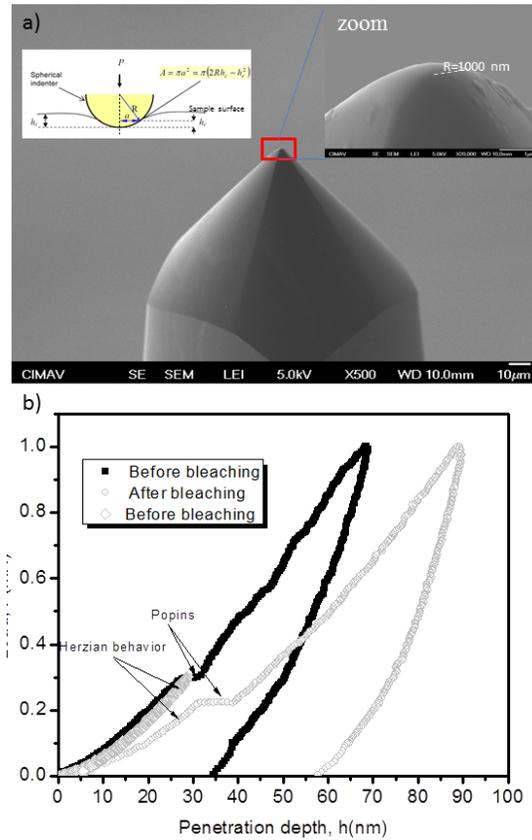


Fig. 7.- a) Optical image, b) and c) AFM images in contact mode taken before and after indentation test with Berkovich tip.

On other works, enamel exhibited a recovery of properties when it was stored in artificial saliva [6], [11], [21]. Unlike on the present work specimens' were stored at dry environment and the enamel did not receive the mineralization benefit of saliva minerals.

#### 4. Conclusions

The hardness values were significantly decreased beyond that 60% as effects of bleaching solution. Variations on elastic modulus values observed in treated specimens' with decrement averages over the 50%. Scan electron microscopy, X-Ray diffraction and Atomic force microscopy analysis evidenced erosive roughness and levels of calcium diminished. This results as whole evidence, exhibited the nanomechanical behavior of great dearest on hardness, elastic modulus and stiffness after bleaching treatment of 35% HP on dehydrated enamel surface.

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